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Determination of the effective sample thickness via radiative capture

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Abstract

A procedure for determining the effective thickness of non-uniform irregular-shaped samples via radiative capture is described. In this technique, partial γ -ray production cross sections of a compound nucleus produced in a neutron-capture reaction are measured using Prompt Gamma Activation Analysis and compared to their corresponding standardized absolute values. For the low-energy transitions, the measured cross sections are lower than their standard values due to significant photoelectric absorption of the γ rays within the bulk-sample volume itself. Using standard theoretical techniques, the amount of γ -ray self absorption and neutron self shielding can then be calculated by iteratively varying the sample thickness until the observed cross sections converge with the known standards. The overall attenuation, thus, provides a measure of the effective sample thickness illuminated by the neutron beam. This procedure is illustrated through radiative neutron capture using a powdered sample comprising enriched ^{186}W from which the effective thickness of the sample is deduced to be 0.077(3) mm.

Keywords: Prompt Gamma Activation Analysis (PGAA), γ -ray absorption, neutron attenuation, (n, γ) , radiative capture, partial γ -ray production cross sections, effective sample thickness.

1. Introduction

There are many applications of in-beam measurements where the irradiated sample may take an irregular inhomogeneous form. For irradiated powders in particular, the material will need to be held in a sample holder such as a Teflon bag. Consequently, the sample mass is unlikely to distribute itself into a regular shape with a uniform surface; measuring the areal density and thickness, or the average pile density, of these non-uniform samples then becomes extremely challenging. For these samples, accurately quantifying the induced radioactive decay, for example, from deexcitation via γ -ray emission, may become complicated. This is especially true for high-density materials with low-to-moderate neutron-absorption cross sections, for example, tungsten, since neutron self-shielding and γ -ray absorption within the sample itself is significant and large corrections are needed. Regular geometries may be modeled rather well using particle-transport simulation toolkits such as MCNP6 [1] and GEANT4 [2]. However, insufficient knowledge of the geometric distribution of the sample, i.e. its thickness, makes it difficult to correct for attenuation in a robust and reliable manner and so alternative procedures must be developed. Some examples of methodologies that have previously been adopted to address this problem include: (i) Measuring neutron-fluence rates in Prompt Gamma Activation Analysis (PGAA), where the irradiated sample is held *in* and *out* of beam, to determine relative self-shielding factors and establish the average pile density of the sample [3]; (ii) Comparing full-energy

photopeaks from ideally-thin calibration standards to the same transitions from *thick* samples to establish the γ -ray energy-dependent (E_γ) correction factors for neutron self shielding and photoattenuation [4]; (iii) Semi-quantitative Monte Carlo simulations to reproduce measured γ -ray intensity maps by iteratively adjusting neutron self absorption and photoattenuation correction factors to minimize residuals between measurement and simulation [5].

Indeed, if self-attenuation processes are not handled accurately, this can lead to misleading partial γ -ray production cross sections (σ_γ) for samples measured using PGAA setups [6, 7]. This problem was highlighted recently in a study of the tungsten isotopes [8] where the new σ_γ values were found to differ considerably from those in the current Evaluated Gamma-ray Activation File (EGAF) [9, 10] repository owing to neutron-absorption and γ -ray self-attenuation considerations. In this paper, we present a method for determining the *effective thickness* of a non-uniform sample via radiative capture through exposure to a flux of near-thermal neutrons. Here, we define the effective thickness as the mean thickness *seen* by the neutrons and γ rays in an equivalent planar geometry corresponding to a sample of the same material. As a validation of the present methodology, we also compare our results to those obtained using a suitably-thin reference calibration standard, similar to the procedure outlined in Ref. [4].

2. Method

A series of *near-thermal* ($T = 120$ K) neutron-capture measurements using isotopically-enriched and natural samples of

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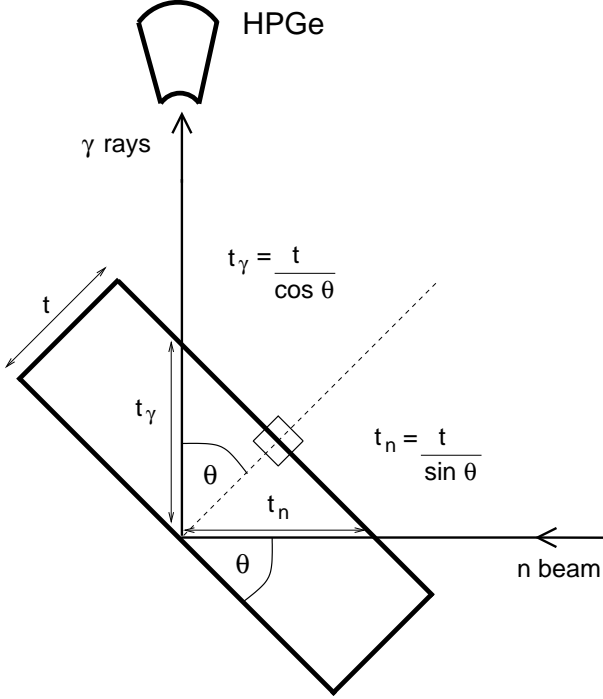


Figure 1: Schematic illustrating the basic setup of the PGAA facility at the BRR. The sample, represented as a rectangle of thickness t , is at an angle $\theta = 30^\circ$ relative to the incident neutron-beam direction and viewed by a High-Purity Germanium (HPGe) detector mounted perpendicular to the beam direction and located ~ 23.5 cm from the sample. The overall exponential attenuation factor may be expressed as a linear combination of axis-resolved γ -ray and neutron-attenuation coefficients: $\exp[-(\mu_\gamma t_\gamma + \mu_n t_n)] = \exp\left[-\left(\frac{\mu_\gamma}{\cos \theta} + \frac{\mu_n}{\sin \theta}\right)t\right]$.

tungsten oxide (WO_2) powders have recently been carried out at the 10-MW Budapest Research Reactor (BRR) [11, 12, 13]. This work employed the PGAA setup [6, 7] to measure absolute neutron-capture partial γ -ray production cross sections in tungsten isotopes produced via the radiative (n, γ) reaction with samples of enriched $^{182,183,186}\text{W}$ and $^{\text{nat}}\text{W}$. Prompt γ -ray spectra were collected and analyzed offline using the γ -ray spectroscopy software package HYPERMET-PC [14]. Full details of the experimental setup and corresponding results are published elsewhere [8]. To obtain these precise cross sections, however, requires accurate knowledge of the *effective* sample thickness exposed to the neutron beam during irradiation. Samples that are not ideally thin, such as the high-density tungsten samples ($\rho(\text{W}) = 19.25 \text{ g/cm}^3$, $\rho(\text{WO}_2) = 10.8 \text{ g/cm}^3$) used in this work, are not fully transparent to neutrons and γ rays. Furthermore, because the distribution of the neutron beam is generally not spatially uniform, and in practice the beam aperture is often smaller than the actual volume of the bulk sample material, the irradiated sample may not be fully illuminated by the neutron beam. Additionally, the partial illumination zone of the sample may not be in full view of a collimated γ -ray detector. These considerations may, thus, contribute significantly to the effective sample thickness deduced from the measured peak areas in PGAA, appropriately corrected for neutron self shielding and γ -ray self absorption due to attenuation within the observed illumination zone of the sample itself.

In Sect. 2.1 and 2.2, respectively, we first outline the theoretical formalism governing the self-attenuation processes, and then describe how this information can be used to experimentally infer the effective thickness of the irradiated sample.

2.1. Self-attenuation correction

In general, scattered photons will emerge from an irradiated sample with a diminished photon intensity I according to the exponential-attenuation law

$$I = I_0 e^{-\mu x}, \quad (1)$$

where I_0 represents the *true* unattenuated photon intensity, x is the sample thickness, and μ is a coefficient describing the linear combination of neutron- (μ_n) and γ -ray (μ_γ) attenuation coefficients for a given sample i.e. $\mu = \mu_\gamma + \mu_n$. The angle θ of the sample holder relative to the incident beam direction, depicted by the geometry in Fig. 1, must also be taken into account in determining the overall attenuation coefficient, thus

$$\mu = \frac{\mu_\gamma}{\cos \theta} + \frac{\mu_n}{\sin \theta}. \quad (2)$$

This angle has been measured to be 30° at the BRR. The analytic form of the γ -ray energy-dependent attenuation factors [15] may then be obtained from the numerical integration of Eq. 1 over the sample thickness t

$$\begin{aligned} \frac{I}{I_0} &= \int_{x=0}^{x=t} e^{-\left(\frac{\mu_\gamma}{\cos \theta} + \frac{\mu_n}{\sin \theta}\right)x} \cdot dx \\ &= \frac{1}{\left(\frac{\mu_\gamma}{\cos \theta} + \frac{\mu_n}{\sin \theta}\right)} \cdot \left[1 - e^{-\left(\frac{\mu_\gamma}{\cos \theta} + \frac{\mu_n}{\sin \theta}\right)t}\right]. \end{aligned} \quad (3)$$

In this work, we derived the γ -ray energy dependent μ_γ values from XMuDat [16], a database of mass-attenuation coefficients, μ_γ/ρ , generated according to prescriptions outlined in Refs. [17, 18]. Thus, for a natural sample of mono-elemental composition $\mu_\gamma = (\mu_\gamma/\rho)\rho$, where ρ is the sample density. For compounds, however, we must introduce a weighting factor w_i to account for the constituent elements in the compound

$$w_i = \frac{A_i}{M} \quad \text{where} \quad M = \sum_i^j A_i. \quad (4)$$

Here, A_i is atomic mass of element i and M is the total stoichiometric mass of the compound. The γ attenuation coefficients for a compound sample may then be deduced using the expression

$$\mu_{\gamma_x} = \left(\sum_i^j \frac{\mu_{\gamma_i}}{\rho_i} w_i \right) \rho_x, \quad (5)$$

where ρ_x is the measured density of the compound sample, and the summation is performed over all constituent elements i in

Table 1: Transition energies (E_γ) in ^{187}W and their corresponding absolute neutron-capture partial γ -ray production cross sections (σ_γ^S) obtained from a standardized $\text{H}_2\text{WO}_4(n, \gamma)$ measurement [8]. The transitions are indexed (i) in order of ascending E_γ . The efficiency-corrected peak areas, denoted A'_γ , were measured using an enriched ^{186}W powdered sample; A_γ^c represent the same peak areas corrected for γ -ray absorption and neutron self shielding assuming the deduced effective sample thickness of 0.077 mm. In the final column the proportionality constant, $R_i(E_{\gamma_i})$, relating σ_γ^S and A_γ^c is listed. See text for details.

E_γ [keV]	i	σ_γ^S [b]	A'_γ [cnt]	A_γ^c [cnt]	$R_i(E_{\gamma_i})$ [b·cnt $^{-1}$]
77.39(3)	1	0.234(4)	13361(321)	25950(623)	$9.02 \times 10^{-6}(27)$
145.79(3)	2	1.344(13)	124860(5619)	146941(6612)	$9.15 \times 10^{-6}(42)$
273.10(5)	3	0.380(4)	38868(3848)	41078(4066)	$9.25 \times 10^{-6}(92)$
5261.68(6)	4	0.653(9)	70901(1205)	72449(1231)	$9.01 \times 10^{-6}(20)$

the compound. The authors of the XMuDat database claim a precision of 5% for the absorption data, so this was adopted as a conservative estimate to derive Δ_{μ_γ} , the uncertainty in μ_γ . For thermal-neutron temperatures and lower, the neutron-attenuation coefficients may be computed from the neutron-absorption cross sections σ_{nabs} listed in Ref. [19] according to

$$\mu_{n_x} = N_A b \frac{T_0}{T} \rho_x \sum_i^j \frac{\sigma_{\text{nabs},i}}{A_i} w_i, \quad (6)$$

where $N_A = 6.022 \times 10^{23}$ is the Avogadro constant, the conversion factor $b = 1 \times 10^{-24} \text{ cm}^2$, $T_0 = 293 \text{ K}$ represents the true thermal-neutron temperature, $T = 120 \text{ K}$ is the assumed near-thermal temperature of the BRR neutron beam. Clearly, for samples of mono-elemental composition $w_i = 1$, in Eq. 6 above, and ρ_x is simply replaced by ρ_i . The uncertainty in μ_n , written as Δ_{μ_n} , is determined by propagating through the uncertainty on σ_{nabs} alone, with all other terms in Eq. 6 treated as constants.

Assuming μ_γ and μ_n to be independent variables, the overall uncertainty on the attenuation factor I/I_0 is then derived using the law of combination of errors

$$\Delta_{I/I_0}^2 = \left[\frac{\partial}{\partial \mu_\gamma} \left(\frac{I}{I_0} \right) \right]^2 \Delta_{\mu_\gamma}^2 + \left[\frac{\partial}{\partial \mu_n} \left(\frac{I}{I_0} \right) \right]^2 \Delta_{\mu_n}^2. \quad (7)$$

Appropriate error propagation and differentiating I/I_0 (Eq. 3) with respect to μ_γ and μ_n , independently, yields the following expression quantifying the variance in I/I_0 from which the overall uncertainty may be deduced

$$\Delta_{I/I_0}^2 = \left[\frac{1}{\frac{\mu_\gamma}{\cos \theta} + \frac{\mu_n}{\sin \theta}} \left(te^{-\left(\frac{\mu_\gamma}{\cos \theta} + \frac{\mu_n}{\sin \theta} \right) t} - \frac{I}{I_0} \right) \right]^2 \times \left(\frac{\Delta_{\mu_\gamma}^2}{\cos^2 \theta} + \frac{\Delta_{\mu_n}^2}{\sin^2 \theta} \right). \quad (8)$$

2.2. Standardization and determination of the effective sample thickness

To establish the effective target thickness of the thick high-density tungsten sample measured in this work, a set of absolute

neutron-capture partial γ -ray production cross sections is required for comparison. Adopting comparator γ -ray transitions, whose cross sections are well known, permits use of an internal-standardization procedure, outlined in detail in Ref. [20]. In this procedure, the observed γ -ray intensities may be normalized by scaling to the well-known comparator transitions observed in the same capture- γ spectrum. For comparative purposes here, we used strong γ lines in ^{187}W that were extracted from a previous thin-sample measurement of a low-density hydrated tungstic acid (H_2WO_4) compound of natural composition [8]. In that measurement, the absolute cross sections for the ^{187}W γ lines, listed in Table 1, were extracted via internal standardization to the 2223-keV transition in hydrogen which has a well-known partial γ -ray production cross section $\sigma_\gamma(2223 \text{ keV}) = 0.3326(7) \text{ b}$ [20], and taking into account the stoichiometric ratio of H : W. The lack of any strong s -wave resonances near the thermal-neutron point at 25.3 meV in tungsten renders applicable a pure $1/v$ dependence on the observed cross section near and below thermal-neutron energies (E_n). Under these circumstances, the cross section may be represented as shown in Ref. [21], accordingly:

$$\sigma_\gamma(E_n; v) = \sigma_0 \sqrt{\frac{E_0}{E_n}} = \sigma_0 \left(\frac{v_0}{v} \right), \quad (9)$$

where σ_0 is the total radiative thermal neutron-capture cross section, $v_0 = 2200 \text{ m/s}$ is the thermal neutron velocity, and $E_0 = 25.3 \text{ meV}$. It follows, therefore, that

$$\sigma_\gamma(E_n) \propto \frac{1}{v} \quad \forall \quad E_n \lesssim 25.3 \text{ meV}. \quad (10)$$

Also, since $(T_0/T) = \sqrt{(E_0/E)}$, at our assumed average beam temperature $T = 120 \text{ K}$, the $1/v$ law implies a corresponding neutron-beam energy centered around 4.2 meV. Thus, although Eqs. 9 and 10 are rigorously defined for monoenergetic neutrons [21], our measurement concerns a distribution of neutrons centered on $\sim 4.2 \text{ meV}$. However, because both tungsten and hydrogen obey the $1/v$ law (i.e. Eqs. 9 and 10 are valid), moreover $\sigma_\gamma(\text{W})$ are deduced relative to $\sigma_\gamma(\text{H})$, it follows that any correction for the neutron-beam energy cancels.

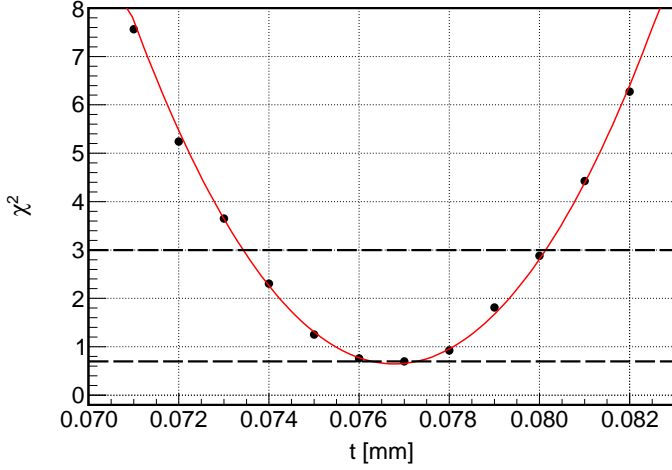


Figure 2: Plot showing the global χ^2 distribution as a function of thickness (t) for the ^{186}W sample irradiated in this work. Dashed lines are drawn to indicate the minimum ($\chi^2_{\min} \approx 0.70$) of the distribution and at $\chi^2_{\min} + 2.3$ to respectively illustrate the effective target thickness t and its uncertainty Δt .

The absolute cross sections, σ_γ^S , of Table 1 [8] were adopted for standardization purposes since they cover the complete energy range of interest: from below 100 keV, where γ -ray self absorption is significant, to several MeV, where absorption is negligible. In general, the measured peak areas A_γ observed in the capture- γ spectrum are related to their corresponding transition cross sections according to the expression

$$A_\gamma = \sigma_\gamma \frac{mN_A}{M} \theta \epsilon_\gamma(E_\gamma) \frac{I(E_\gamma)}{I_0} \phi \tau T, \quad (11)$$

where m (the sample mass), N_A , and M are described in Sect. 2.1, θ is the isotopic abundance, ϕ is the neutron flux, τ is a data-acquisition-deadtime correction factor, T is the irradiation period, and $\epsilon_\gamma(E_\gamma)$ denotes the relative γ -ray energy-dependent detection efficiency. The detection efficiency represents a total efficiency given by the product of the intrinsic-detection efficiency and the geometric efficiency describing the solid angle subtended by the detector with respect to the sample position. Because m , N_A , M , θ , ϕ , τ , and T are systematic quantities that contribute to the observed count rate in all γ lines, collectively they constitute a constant K , such that

$$\frac{A'_\gamma}{\sigma_\gamma} \cdot \frac{1}{I(E_\gamma)/I_0} = K, \quad (12)$$

where $A'_\gamma = A_\gamma/\epsilon_\gamma(E_\gamma)$ is the efficiency-corrected peak area. Although both ϵ_γ and the attenuation factor I/I_0 are dependent on E_γ , since the measured peak area for a given transition is directly proportional to its corresponding absolute standard cross section, it follows that the ratio of σ_γ^S to the measured peak area will be constant for any transition provided the peak areas have been appropriately corrected for these quantities. This condition may be expressed as

$$\frac{\sigma_{\gamma_1}^S(E_{\gamma_1})}{A_{\gamma_1}^c(E_{\gamma_1})} = \frac{\sigma_{\gamma_2}^S(E_{\gamma_2})}{A_{\gamma_2}^c(E_{\gamma_2})} = \frac{\sigma_{\gamma_3}^S(E_{\gamma_3})}{A_{\gamma_3}^c(E_{\gamma_3})} = \dots, \quad (13)$$

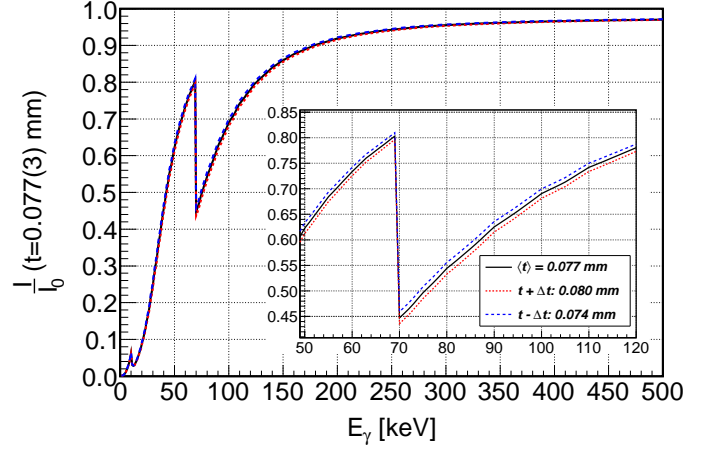


Figure 3: The γ -ray energy-dependent attenuation factor, I/I_0 , plotted as a function of E_γ for the ^{186}W sample assuming $t = 0.077(3)$ mm in Eq. 3. The curve is expanded around $E_\gamma = 50$ -120 keV to reveal the upper and lower bounds of the uncertainty band (for one standard deviation). This curve provides an E_γ -dependent correction to the measured transition intensities in the capture- γ spectrum. The abrupt K -edge in tungsten is clearly visible at around 69-70 keV and corresponds to a sharp increase in μ_γ at the K -shell electron binding energy; the smaller cumulative effect of the L -I, L -II, and L -III edges is also observed in the low-energy regime spanning approximately 10-12 keV.

or, in general terms,

$$R_i(E_{\gamma_i}) = \frac{\sigma_{\gamma_i}^S(E_{\gamma_i})}{A_{\gamma_i}^c} = \text{const.}, \quad (14)$$

where $A_{\gamma_i}^c$ represents the attenuation-corrected measured peak area given by

$$A_{\gamma_i}^c = \frac{A'_\gamma}{I/I_0(t)}, \quad (15)$$

and A'_γ is the ϵ_γ -corrected attenuated peak area measured directly in the capture- γ spectrum. The effective sample thickness is then determined by treating t as an adjustable parameter in Eq. 3 and varying this quantity until the condition embodied by the ratio in Eqs. 13 and 14 converges upon a unique result. An appropriate energy-dependent attenuation correction can then be applied to the entire capture- γ spectrum.

Since the ratios $R_i(E_{\gamma_i})$ are not independent measurements (each with uncertainty ΔR_i) but are in fact correlated, a covariance matrix \mathbf{V} is needed to describe them. From the principle of maximum likelihood we know that the probability of a particular set of data is the product of the individual probabilities, and this provides a basis for the least squares method adopted in this analysis. By minimizing the exponent in the Gaussian likelihood function we may determine the global χ^2 for an assumed effective thickness t using the following expression

$$\chi^2 = \sum_{i=1}^N \sum_{j=1}^N [R_w - R_i(E_{\gamma_i})][V_{ij}^{-1}][R_w - R_j(E_{\gamma_j})], \quad (16)$$

where R_w is the expectation value at a given t determined from the weighted average of the corresponding set of $R_i(E_{\gamma_i})$ measurements. The individual (i, j) elements of an $N \times N$ covariance matrix, where N represents the maximum number of data

Table 2: Standard cross sections in ^{187}W , σ_γ^S , are taken from the H_2WO_4 normalization measurement [8] and the attenuated ^{187}W cross sections, σ_γ^T , are from the thick-sample measurement using enriched ^{186}W . Cross sections are corrected for isotopic abundance of ^{186}W [23]. The experimental attenuation factors $(I/I_0)_{\text{exp}}$ were deduced from the ratio $\sigma_\gamma^T/\sigma_\gamma^S$ (Eq. 19) and compared to the theoretical values I/I_0 (Eq. 3) for an effective sample thickness $t = 0.077$ mm, assuming: (1) both γ -ray absorption and neutron self shielding; (2) γ -ray absorption only. Uncertainties are smaller in the first case owing to the overall non-linearity represented by the combination μ_γ and μ_n according to the functional form of Eq. 8.

E_γ [keV]	σ_γ^S [b]	σ_γ^T [b]	$(\frac{I}{I_0})_{\text{exp}}$	$(\frac{I}{I_0})_{t=0.077 \text{ mm}}^1$	$(\frac{I}{I_0})_{t=0.077 \text{ mm}}^2$
77.39(3)	0.823(14)	0.433(14)	0.526(19)	0.514(25)	0.522(26)
145.79(3)	4.727(46)	4.05(20)	0.856(44)	0.849(37)	0.867(43)
273.10(5)	1.337(14)	1.26(13)	0.942(96)	0.943(30)	0.963(48)
5261.68(6)	2.297(32)	2.297(64)	1.000(31)	0.975(15)	0.997(49)

points, are given as $V_{ij} = \Delta_{R_i}^2 \delta_{ij}$ for the diagonal elements (where the Kronecker delta function $\delta_{ij} = 1$ for $i = j$, and 0 for $i \neq j$) and $V_{ij} = r_{ij} \Delta_{R_i} \Delta_{R_j}$ for the off-diagonal matrix elements, where r_{ij} is the correlation coefficient between a pair of $R_i(E_{\gamma_i})$ measurements. For each of the N data points the corresponding ratios $R_1(E_{\gamma_1}), R_2(E_{\gamma_2}), \dots, R_N(E_{\gamma_N})$, may be denoted \mathbf{R} , where \mathbf{R} is a vector of N elements. Writing \mathbf{R}_w also as an N -element vector then allows Eq. 16 to be conveniently recast in matrix notation as

$$\chi^2 = (\mathbf{R}_w - \mathbf{R})\mathbf{V}^{-1}(\widetilde{\mathbf{R}}_w - \widetilde{\mathbf{R}}), \quad (17)$$

where $\widetilde{\mathbf{R}}_w$ and $\widetilde{\mathbf{R}}$ represent the transposed vectors \mathbf{R}_w and \mathbf{R} , respectively.

In this work, two parameters (P) are adjusted to minimize χ^2 to fit the $N = 4$ data points: t and r_{ij} . Here, r_{ij} represents an average solution. The number of degrees of freedom (ndf) is thus, $N - P = 2$. The reduced χ^2 , i.e. χ^2/ndf , for a fit with $\text{ndf} = 2$ is expected to be ~ 0.35 at the 68.3% confidence level [22]. The corresponding value of t yielding a global $\chi_{\min}^2 \approx 0.7$ for the fit from a two-parameter adjustment may then be interpreted as the mean effective sample thickness, and its uncertainty Δt extracted from the range of values for $t \pm \Delta t$ satisfying the condition $\chi^2 \leq \chi_{\min}^2 + 2.3$ [22]. Accordingly, we estimate Δt using

$$\Delta t = \frac{1}{2}[t_{>}(\chi_{\min}^2 + 2.3) - t_{<}(\chi_{\min}^2 + 2.3)], \quad (18)$$

where $t_{>}$ and $t_{<}$ are the interpolated values of t at $\chi_{\min}^2 + 2.3$, with $t_{>} > t$ and $t_{<} < t$.

3. Results

To illustrate the procedure for determination of the effective thickness t , we present results from an (n, γ) measurement for a 169-mg powder sample of $^{186}\text{WO}_2$ (99.65(3)% enrichment) irradiated for a period of 2.03 h to induce prompt radioactivity in the ^{187}W compound nucleus [8]. Alongside the standardization-transition energies and cross sections listed in Table 1 are the corresponding ϵ_γ -corrected attenuated peak areas, A'_γ . Using

this data, we iteratively step through values of t over a suitable range of thicknesses to calculate $A_{\gamma_i}^c$ (Eq. 15) to find a statistically consistent $R_i(E_{\gamma_i})$ (Eq. 14) data set. Simultaneously, we adjust r_{ij} between 0 and 1 to minimize the χ^2 function in Eq. 17. A tungsten effective sample thickness $t = 0.077$ mm (equivalent to 0.16 mm for WO_2) assuming an overall average correlation coefficient $r_{ij} = 0.82$ can reproduce $\chi_{\min}^2 = 0.70$ i.e. the global χ^2 expectation value for a minimization based on $\text{ndf} = 2$. The corresponding results for $A_{\gamma_i}^c$ and $R_i(E_{\gamma_i})$ at $t = 0.077$ mm are also summarized in Table 1, from which an average $R_w = 9.04 \times 10^{-6} \text{ b}\cdot\text{cm}^{-1}$ is determined. The elements of the covariance matrix supporting this result may be written as a product of the correlation coefficient and the associated $R_i(E_{\gamma_i})$ uncertainty products, Δ_{R_i} from Table 1. Because $N = 4$, the general form of the 4×4 covariance matrix may be represented as

$$V = \begin{pmatrix} r_{11} \Delta_{R_1} \Delta_{R_1} & r_{12} \Delta_{R_1} \Delta_{R_2} & r_{13} \Delta_{R_1} \Delta_{R_3} & r_{14} \Delta_{R_1} \Delta_{R_4} \\ r_{21} \Delta_{R_2} \Delta_{R_1} & r_{22} \Delta_{R_2} \Delta_{R_2} & r_{23} \Delta_{R_2} \Delta_{R_3} & r_{24} \Delta_{R_2} \Delta_{R_4} \\ r_{31} \Delta_{R_3} \Delta_{R_1} & r_{32} \Delta_{R_3} \Delta_{R_2} & r_{33} \Delta_{R_3} \Delta_{R_3} & r_{34} \Delta_{R_3} \Delta_{R_4} \\ r_{41} \Delta_{R_4} \Delta_{R_1} & r_{42} \Delta_{R_4} \Delta_{R_2} & r_{43} \Delta_{R_4} \Delta_{R_3} & r_{44} \Delta_{R_4} \Delta_{R_4} \end{pmatrix},$$

where, for the diagonal matrix elements $r_{ii} = 1$ (see Sect. 2.2), and for all off-diagonal elements $r_{ij} = 0.82$. Thus, using these values for r_{ij} together with the corresponding indexed uncertainties from the final column of Table 1 yields

$$V = \begin{pmatrix} [1.00 \times 0.27 \times 0.27] & \cdots & [0.82 \times 0.27 \times 0.20] \\ [0.82 \times 0.42 \times 0.27] & \cdots & [0.82 \times 0.42 \times 0.20] \\ [0.82 \times 0.92 \times 0.27] & \cdots & [0.82 \times 0.92 \times 0.20] \\ [0.82 \times 0.20 \times 0.27] & \cdots & [1.00 \times 0.20 \times 0.20] \end{pmatrix}.$$

For clarity, we have factored through the 10^{-6} common denominator associated with all $R_i(E_{\gamma_i})$ and Δ_{R_i} values of Table 1, thus simplifying the χ^2 calculation.

The overall uncertainty on the target thickness is determined by plotting the global χ^2 distribution as a function of t and interpolating the range of t consistent with $\chi^2 \leq \chi_{\min}^2 + 2.3 \lesssim 3$.

Figure 2 shows that values of t between approximately 0.0734 and 0.0801 mm (3 sf) satisfy this criterion window of $\chi^2 \lesssim 3$. Hence, from Eq. 18 we report an estimate for $\Delta t = \pm 0.003$ mm (3 dp). Our result for the *effective target thickness* of the inhomogeneous powder sample, therefore, describes the equivalent planar thickness of the tungsten sample based on the measured apparent overall self attenuation within the sample. The one-standard-deviation γ -ray energy-dependent attenuation correction, applicable to the observed γ -ray intensities measured in the (n, γ) capture spectrum [8], corresponding to an effective thickness $t = 0.077(3)$ mm is shown in Fig. 3.

As a consistency check, the overall attenuation factors may also be deduced experimentally by comparing the *attenuated* partial γ -ray production cross sections from the *thick* high-density ^{186}W measurement, referred to as σ_γ^T , to the same transitions from the reference standard with known absolute cross sections, σ_γ^S . A similar procedure is described in Ref. [4]. The attenuated σ_γ^T values are also deduced via the internal-standardization procedure (see Sect. 2.2 and references therein); in this instance by scaling to the high-energy comparator line in ^{187}W at 5261.68 keV which is subject to negligible γ -ray absorption. Because the thick-sample cross sections, σ_γ^T , have not been corrected for self attenuation, and σ_γ^S represents the ideal standard for a given transition, the experimental attenuation factor, $(I/I_0)_{\text{exp}}$, may be defined by the following ratio

$$\left(\frac{I}{I_0}\right)_{\text{exp}} = \frac{\sigma_\gamma^T}{\sigma_\gamma^S}. \quad (19)$$

Clearly, for lower-energy transitions the observed attenuation is more apparent and although $\sigma_\gamma^T < \sigma_\gamma^S$ for all transitions, σ_γ^T asymptotically reaches σ_γ^S as E_γ increases beyond $\gtrsim 300$ keV (where $I/I_0 \rightarrow 1$). Presented in Table 2 are the standardized thick-sample cross sections, corrected for the 28.43% isotopic abundance of ^{186}W [23], and the corresponding experimental attenuation factors. The effective sample thickness extracted from the covariance analysis may then be compared to the observed experimental values given by Eq. 19 for each E_γ . As shown in Table 2, the theoretical values for I/I_0 for a $t = 0.077$ -mm sample thickness compare favourably with the experimentally-deduced values. Remarkable consistency is achieved assuming: (i) both γ -ray self absorption and neutron self shielding (Eq. 3); (ii) γ -ray self absorption alone (Eq. 3 without the μ_n dependence). In the thick-sample approximation, the irradiated sample absorbs all incident neutrons allowing the γ -ray absorption to be treated independently. Our results in this case reveal very little difference in the calculation of $I/I_0(\mu_\gamma, \mu_n)$ and $I/I_0(\mu_\gamma)$ for $t = 0.077$ mm and both sets of calculations are in excellent agreement with the experimentally-deduced attenuations. This agreement is reinforced in Fig. 4 which shows excellent consistency between calculated and experimental attenuation factors, not only for the transitions used in deriving the effective sample thickness (red data points), but also for other prominent transitions (black data points) compared in the standard (H_2WO_4) and enriched $^{186}\text{W}(n, \gamma)$ spectra. This observation provides a useful verification of the adopted method for the determination of the effective sample thickness.

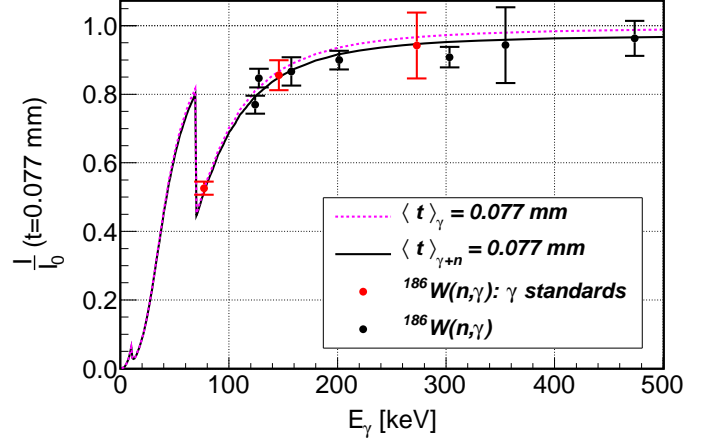


Figure 4: The γ -ray energy-dependent attenuation factor, I/I_0 , plotted as a function of E_γ for energies up to 500 keV, assuming $t = 0.077$ mm. The solid-black curve corresponds to $I/I_0(\mu_\gamma, \mu_n)$ and the dashed-magenta curve neglects the μ_n dependence, thus corresponds to $I/I_0(\mu_\gamma)$. The data points show the experimentally-deduced attenuation factors for individual γ -ray transitions using Eq. 19: the red points correspond to transitions used to deduce the sample thickness; the black points are representative transitions from the $^{186}\text{W}(n, \gamma)$ spectra.

4. Conclusion

A capability has been demonstrated to precisely measure the effective thickness of a sample with an irregular geometry via radiative capture. This technique is predicated on quantifying the γ -ray absorption observed in low-energy transitions by comparison to known absolute partial γ -ray production cross sections, and has been experimentally verified through neutron-capture measurements performed with the PGAA setup at the BRR. We have successfully illustrated this concept to a precision of better than 4% for a 0.077(3)-mm measurement of the tungsten-equivalent effective sample thickness for a WO_2 powdered sample.

For materials, such as tungsten, with low-to-moderate neutron-capture cross sections, neutrons are more likely to penetrate deeper into the sample, and so radioactivity is likely to be induced from deeper within the bulk-sample volume. Consequently, the transmission path through which the γ rays must traverse, on average, is likely to represent the mean thickness of the irradiated sample. Because tungsten is also a high- ρ density material, the probability of complete photoelectric absorption within the sample is also high leading to large corrections, particularly for the lower-energy ($\lesssim 300$ keV) transitions that are subject to higher degrees of γ -ray self absorption. The amount of overall attenuation, therefore, provides a measure of the apparent, effective, thickness of the sample volume illuminated by the neutron beam. In general, this measured effective thickness is likely to be somewhat less than the *true* thickness represented by the entirety of the bulk sample material, and the amount of absorption and self shielding from an integration over the true thickness may constitute an over estimate of the correction factor. Furthermore, γ -ray transmission through samples with high neutron-capture cross sections, or low- ρ , is far less likely to be impeded to a measurable extent and this represents a limitation of the technique.

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